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Preparation and Properties of Transition Metal Beta" Aluminas

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## PREPARATION AND PROPERTIES OF TRANSITION METAL 6" ALUMINAS

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Ion exchange techniques were used to produce a family of 3d transition metal  $\beta'$ -aluminas by substitution of the sodium ion content in  $Na^+-\beta'$ -alumina single crystals. Complete or nearly complete replacement was obtained for  $Cr^{3+}$ ,  $Nn^{2+}$ ,  $Co^{2+}$ , and  $Ni^{2+}$  ecropositions. Optical properties of the crystals were studied and both broad absorption and broad fluorescence bands were observed. Environmental exposure produced detrimental effects over long time periods, suggesting hydration of the conduction planes.

### 1. INTRODUCTION

Previous studies<sup>1,2</sup> have found that the entire sodium ion content of β'-alumina can readily be replaced by a variety of divalent and trivalent cations. In several cases, it has been found that the resultant materials have interesting optical properties in addition to fast ion transport. For example, Hd<sup>3+</sup> exchanged β'-alumina single crystals have been found to lase and to exhibit extremely high fluorescent gain coefficients in comparison with Hd:YAG lasers.<sup>3</sup>

The research on lanthanide \$'-aluminas prompted our interest in the 3d-transition metals as candidates for exchange into \$'-alumina. Several ions within this series exhibit broadband, vibronic fluorescent transitions due to the interaction of their incomplete 3d electron shells with local crystal fields. This phenomenon allows for the development of tunable solid state lasers using various exide and non-exide host materials. Relatively little work has been done, however, with \$'-alumina systems

in which the exchanged ions were capable of multiple valence states. In addition to this problem, the exchange of transition metal ions generally involves the substitution of ions with very small radii, potentially inducing large stresses within the crystals. The preparation of these materials, and their interesting optical properties are reported in this paper.

### 2. PREPARATION OF THE MATERIALS

Several transition metal  $\beta'$ -aluminas were prepared using standard ion exchange techniques on single crystals of Na<sup>+</sup>- $\beta'$ -alumina (Table 1). Crystals were either immersed in molten halide salts of appropriate compositions or placed in the proximity of heated, but unmelted, salts for varying times. In this latter case the salt exhibited a sufficiently high vapor pressure such that ion exchange occurred from exposure to the vapor phase. For all samples, careful atmospheric control was required to prevent exidation of the halide salts as well as to prevent

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TABLE 1: Ion Exchange Conditions for Transition-Metal-8"-Aluminas

ION	EXCHANGE MEDIUM		TEMPERATURE ( °C)	THE(hr)	PERCENT EXCHANGED
Cr <sup>3+</sup>	Liquid: Vapor:	Sicrcl <sub>3</sub> /69NaCl CrCl <sub>3</sub>	700	1 12	30 90
Mn <sup>2+</sup>	Liquid:	MnCl <sub>2</sub>	650		
co <sup>2+</sup>	Liquid:	CoCl <sub>2</sub>	<b>750</b>	2	100
Wi <sup>2+</sup>	Liquid:	32N1Cl <sub>2</sub> /68NaCl	700	12.5	25
	Vapor:	NiCl <sub>2</sub>		35	90

formation of multiple valence states of the cations.

Results indicate that substantial or complete exchange was attained for  ${\rm Cr}^{3+}$ ,  ${\rm Mn}^{2+}$ .  ${\rm Co}^{2+}$ , and  ${\rm Mi}^{2+}$  using both the liquid and vapor phase techniques. The crystals possess good mechanical integrity and x-ray diffraction indicates that the  $\beta'$ -alumina structure was retained without modification, despite the small size of the substitutional ions. The extent of the exchange reactions was monitored by both gravimetric means and Energy Dispersive X-ray (EDX) analysis, the latter also being used to identify impurities and inhomogeneities in the crystal.

The vapor phase exchanges, achieved by either burying the crystal in an excess of the halide powders or by suspending it above the powder source, required longer reaction times than corresponding melt exchanges. This method, however, enables one to obtain complete exchange at much lower temperatures than those required for the melt phase. In general, the resulting milder synthesis conditions prevent detrimental effects to the crystal. For example, complete exchange of Mi<sup>2+</sup> ions from a melt of MiCl<sub>2</sub> would require melting the salt in excess of 1100°C, while the vapor phase exchange was possible at 700°C. This temperature reduction is critical, for it has been shown that several of the

exchanged  $\beta'$ -aluminas are prone to decomposition when heated to temperatures of above 900°C.<sup>4</sup>

### 3. OPTICAL PROPERTIES

Although most lasers operate at single wavelengths, the success of the room temperature operation of the Cr:BeAl204 (alexandrite) tunable laser, operational over the range for 7300 to \$0001, has spurred much interest in the utilization of broad band vibronic transitions. Most radiative transitions within the 3d transition metal ions take place via such broad bands. and thus, there is the potential for a whole series of solid state lasers that are tunable over large wavelength regions. Numerous studies of transition metal ions in various crystal hosts have taken place. Although there is a wealth of spectroscopic information, there are relatively few effective laser hosts available. Our initial experiments were performed with the intent of determining the applicability of B'alumina as a host for tunable solid state lasers, in that it has already demonstrated success as a laser host for Md3+ ions.

Absorption and fluorescence spectra for several single crystals of ion exchanged  $\beta'^-$  alumina with various  $\text{Cr}^{3+}$  concentrations (the balance being residual  $\text{Ma}^+$  ions within the conduction plane) were performed. The absorption

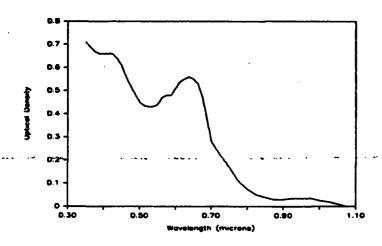


figure 1. Absorption Spectrum of Cr3+-\$'-alumina.

spectra were obtained using a Cary Model 14 Spectrophotometer. Cr3+-8'-alumina was found to have strong absorption peaks centered at 4200A and 6400A (Fig. 1), giving the crystals a deep treen color. Fluorescent emission (induced by numping with a Spectra-Physics argon ion laser it 4880A) was observed from 6900A to 9000A. hese values are quite typical of Cr3+ optical pectra in a variety of other hosts, representng interaction between the  $^4T_1$  (at 4200A) nd 4T, (at 6400A) states with the vibronically roadened ground state 4A2. No evidence of the arity and spin forbidden 2 E + 4A2 transition s observed, probably due to overlap of the <sup>2</sup>E tate with the  ${}^4T_2$ . The placement of these ransitions suggests that the local crystal ield environment of the Cr3+ ions within the "-alumina conduction plane is octahedral, which ould place them within the mid-oxygen (m0) osition of the crystal.

The absence of the  $^2$ E line suggests that he magnitude of the crystal field environment n  $\text{Cr}^{3+}\beta'$ -alumina is similar to that of  $\text{Cr}^{3+}$  in SGG [Gd<sub>3</sub>(Sc,Ga)<sub>2</sub>Ga<sub>3</sub>O<sub>12</sub>], which has an stahedrally symmetric  $\text{Cr}^{3+}$  site with a crystal ield of Dq = 1450 cm<sup>-1</sup> and a Racah parameter = 658 cm<sup>-1</sup> (Dq/B = 2.20). This produces

a direct overlap of the <sup>2</sup>E and <sup>4</sup>T, levels.

However, the observed room temperature fluorescent lifetimes of the  $\beta'$ -alumina crystals is much greater than that observed in GSGG (650 versus 115 microseconds). This trend usually suggests that there is a greater separation between the  $^2$ E level, which typically has lifetimes of several milliseconds, and the  $^4$ T<sub>2</sub> level, whose lifetime is measured in the tens of microseconds. Given that we observe a lifetime greater even than that of alexandrite (262 microseconds at room temperature) $^6$ , it would suggest a larger crystal field than concluded above, more on the order of Dq/B = 2.5.

The temperature dependent behavior of the fluorescent lifetime of  $Cr^{3+}$ - $\beta$ '-alumina fails to provide any explanation for these contradictory observations. The lifetimes are found to decrease exponentially with increasing temperature, inferring the presence of some competing non-radiative process. A plot of the non-radiative decay rate versus inverse temperature (Fig. 2) indicates that the process has an activation energy of 0.16 eV, which is of the same order of that for sodium ion motion in this temperature range. While it is tempting to associate the non-radiative processes with Na<sup>+</sup>

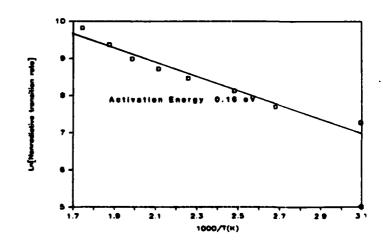
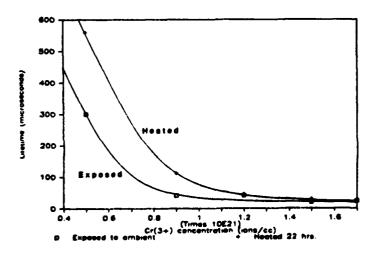


Figure 2. Nog-radiative decay rate in Cr -3'-alumina versus temperature.

ion motion, there may be other competing non-radiative processes, such as multi-phonon decay, and partitioning via Boltzmannn distributions between the closely lying <sup>2</sup>g and <sup>4</sup>T<sub>2</sub> levels. <sup>6</sup> More detailed optical and ion transport studies are necessary in order to understand the non-radiative mechanisms occurring in this solid electrolyte.

In addition to the above effects there also seem to be other processes which contribute to the observed values of fluorescent lifetimes in these crystals. Concentration quenching has been observed, as increasing amounts of Cr3+ ions within the conduction plane have been found to reduce the lifetime by more than an order of magnitude (Fig. 3). Another quenching effect on the fluorescent lifetime occurs from the exposure of the crystals to ambient environments over long periods of time. For example, a crystal whose lifetime immediately after exchange was as high as 2 msec was found to degrade to 300 microseconds after prolonged embient exposure (i.e., months). The process was found to be at least somewhat reversible. Heating the crystal to 350°C for 48 hours raised the lifetime to 650 microseconds.

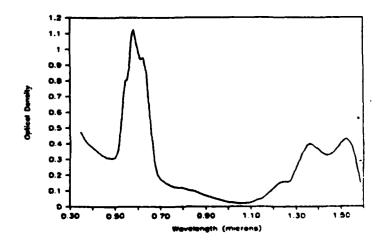
Elevated heating was also found to affect



Pigure 3. Fluorescent lifetime of Cr-5'alumina versus Cr concentration in
hydrated and dried states.

the size of the c-axis lattice parameter of the crystals. In the unheated (short lifetime) state, a c-axis value of \$3.666A was measured, whereas after 48 hours of heating, the value was \$3.547A. The ability of water molecules to diffuse into \$'-alumina and alter their lattice parameters is well established. In addition, water is well known for its ability to quench fluorescence over a wide variety of energies due to its many vibrational and bending modes. The present results indicate that certain optical properties may be sensitive to moisture. A more controlled study of this behavior is necessary to identify and characterize these effects.

The absorption spectra for  $\text{Co}^{2+}$  ion exchanged  $\beta'$ -alumina was also measured (Fig. 4), revealing strong peaks at 5500, 5825, and 6200A, and lesser, but still strong, absorptions at 12,500, 13,625 and 15,250A. The position of these peaks is in excellent agreement with the results reported for  $\text{Co}^{2+}$  in various oxide hosts containing tetrahedral sites. <sup>10</sup> This indicates that the primary location of the  $\text{Co}^{2+}$  ions is the tetrahedral Beevers-Ross type site (6c), despite the fact that most divalent ions in  $\beta'$ -alumina exhibit significant population of both the BR and mid-oxygen sites. <sup>11</sup>



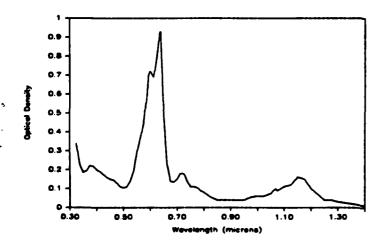
Pigure 4. Absorption Spectrum of Co<sup>2+</sup>-p'alumina.

The excellent agreement with the theoretical line positions, as well as the absence of any unidentified lines helps to verify that all the cobalt ions present in the crystal are in the 2+ state, and that no Co<sup>3+</sup> ions were formed during the exchange process.

Absorption spectra for Ni<sup>2+</sup>-\$'-alumina (Fig. 5) produces peaks at 3760, 6000, 6375, 7200 and 11,500A. This latter absorption is positioned similarly to the absorption due to the first excited state of Ni<sup>2+</sup> in MgO. 12 However, the complexity of the spectra and the sensitivity of the Ni<sup>2+</sup> ion to crystal fields prevents the identification of the other lines in the spectrum. Low temperature measurements will be necessary to resolve some of the overlapping and determine crystal field strength and site symmetry.

4. CONCLUSION

aluminas by the substitution of 3d transition metal ions for Nations has enabled us to begin investigating a new series of materials with potential as solid state tunable lasers. The optical properties of these materials compare favorably with those of materials previously



Pigure 5. Absorption Spectrum of Ni<sup>2+</sup>-s'-alumina.

as a preparation technique permit the synthesis of many samples covering a wide variety of active ion concentrations. Initial results with Cr. exchanged alumina indicate that hydration effects may influence the optical properties of this material.

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### REFERENCES

- G. C. Farrington, B. Dunn and J. O. Thomas, Apl. Phys. A. 32, (1983) 159.
- G. C. Farrington and B. Dunn, Solid State Ionics 7, (1982) 267.
- 3. A. J. Alfrey, O. M. Stafaudd, B. Dunn, D. L. Yang and L. Salmon, submitted for publication.
- 4. D. L. Yang, unpublished results.
- 5. G. Huber and K. Petermann, in Solid State Tunable Lasers, (Springer-Verlag) 1985.
- J. C. Walling, et al., IEEE Journal of Quantum Electronics, QE-16, (1980) 1302.
- 7. G. G. Imbusch in Spectroscopy of Solid-State Laser-Type Materials, B. DiBartolo, ed., in press.
- J. B. Bates, et al., Solid State Ionics 5
   (1981) 159.
- 9. J. B. Bates, et al., Solid State Ionics, 9/10, (1983) 237.
- P. Cossee and A. B. VanArkel, J. Phys. Chem. Solids 15 (1960) 1-6.
- J. O. Thomas, et al., Solid State Ionics 9/10 (1983) 301.
- 12. W. Low, Phys. Rev. 109 (1958) 247.

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